## Comparison of Resistive Switching between Tantalum Oxides Deposited by Atomic Layer Deposition and Ebeam Physical Vapor Evaporation

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Performance of Cu/Ta<sub>x</sub>O<sub>v</sub>/Pt devices is compared between tantalum oxide deposited by atomic layer deposition (ALD) and electron beam physical vapor evaporation (EBPVD). Formation, set and reset of Cu and oxygen vacancy  $(V_0)$  conductive filaments (CF) [1] are evaluated for both dielectrics and related to the respective material properties. XPS data shows that ALD forms a stoichoimetric  $Ta_2O_5$ , while EBPVD results in oxygen deficient TaO<sub>x</sub> films. Forming voltages of Cu and V<sub>o</sub> CFs for ALD Ta<sub>2</sub>O<sub>5</sub> devices are considerably higher than for EBPVD TaOx devices and its  $V_{\text{form,}} \; V_{\text{set}}, \; V_{\text{reset}}$  and  $R_{\text{on}}$ distributions are much tighter than the corresponding distributions for EBPVD TaOx. ALD Ta2O5 devices outperform those with EBPVD TaOx in DC sweeping and in pulse measurements. The resistive Cu/Ta<sub>x</sub>O<sub>y</sub>/Pt devices have been fabricated in a crossbar array on a thermally oxidized Si wafer. Metal electrodes were deposited by e-beam evaporation and patterned by lift-off technology. Both types of devices have the same  $Ta_xO_y$ thickness of 16nm. In our measurements, Pt electrode is grounded and the bias is applied to the Cu electrode.

The improved device performance of ALD devices is attributed to more conformal, uniform, and denser ("pinhole free")  $Ta_2O_5$  thin films formed by ALD than by EBPVD. Defects are known to be essential for resistive memory devices. However, a high level of defects in the dielectric reduces the device reliability [2]. The dielectric constants of both tantalum oxide films are measured by Keithley C-V test.  $Ta_2O_5$  has a higher dielectric constant, about 38, than  $TaO_x$ , around 16, further implying lower level of defects in  $Ta_2O_5$  than in  $TaO_x$ .

For the characterization of oxygen vacancy  $V_o$  CFs a negative voltage is swept from 0 to -8V for the forming and from 0 to -4V for the set operation. At a certain  $V_{form}$ , a forming operation is observed when the cell changes from high resistive state (HRS) to a low resistive state (LRS). Since under negative voltage the migration of Cu<sup>+</sup> ions is suppressed, the LRS is attributed to the formation of a  $V_o$  CF in agreement with [1]. For Cu CF set operation, a positive voltage is swept from 0 to 5V for forming and from 0 to 3V. A compliance current (I<sub>cc</sub>) of 100  $\mu$ A is used for the set operation to avoid damage to the device. A compliance current of 0.1A is used in the reset operation for both  $V_o$  and Cu CFs with a bias voltage sweeping from 0 V to -2 V.

An ALD based device has typically  $V_o$  and Cu CF forming voltages of about -6.5V and 3.5V, respectively, much higher than those of EBPVD devices, which are around -4V and 2V, respectively. After the initial forming, the set voltages for both ALD and EBPVD devices are found to be around -2.5V for Vo CFs and 1.5V for Cu CFs. This is in agreement to PVD samples with different TaO<sub>x</sub> thicknesses of 8nm, 16nm, and 32nm [3]. In the case of EBPVD devices, it was found that V<sub>form</sub> for both Cu and Vo CFs increased considerably with the TaO<sub>x</sub> thickness while V<sub>set</sub> and V<sub>reset</sub> voltages were independent of the TaO<sub>x</sub> thickness. V<sub>set</sub>, V<sub>reset</sub> distributions and R<sub>on</sub> distributions of Cu and Vo CFs are shown in Fig.1 and

Fig.2, respectively. It can be seen that set and reset voltages for Vo CF for ALD and for EBPVD are similar, but the absolute values are more scattered for EBPVD devices, resulting in an overlap of the set and reset ranges. In Fig.2 it is seen that Ron distributions for ALD Ta<sub>2</sub>O<sub>5</sub> are much tighter than the distributions of EBPVD TaO<sub>x</sub>. The variability of Ron is likely related to greater variability of defect densities in EBPVD TaO<sub>x</sub>. The narrow distribution of Ron in ALD devices is to significantly lower defect levels and higher density in Ta<sub>2</sub>O<sub>5</sub> than in EBPVD TaO<sub>x</sub>. It is also observed that ALD devices with Vo switch more reliably than with Cu CF. For ALD devices Ron of Vo CFs generally vary from a hundred ohms to less than  $1K\Omega$ , while the Ron of Cu CFs range from a few hundred ohms to 50KΩ. This wide variation of Ron(Cu-CF) entails wide  $V_{\text{reset}}$  distribution which may interfere with  $V_{\text{set}}$  for Cu CFs. It has been observed that Cu CFs with large  $R_{\rm on}$ display considerable instability leading in some cases to self-dissolution and volatile switching behavior [4]. The retention tests on both devices performed at 85C° show uniform (stable)  $R_{\text{on}}$  and  $R_{\text{off}}$  distribution with retention time up to  $10^{\circ}$  s.

In conclusion, ALD deposition allows for highly conformal and uniform dense stoichoimetric  $Ta_2O_5$  films with a low concentration of defects and high density. Such Cu/Ta<sub>2</sub>O<sub>5</sub>/Pt devices exhibit more reliable switching than EBPVD Cu/TaO<sub>x</sub>/Pt devices with much tighter distribution of the switching parameters. Moreover, the comparison between ALD and EBPVD devices gives insight into the mechanisms leading to forming, set and reset processes of the Cu and V<sub>o</sub> nanofilaments.



Fig.1 V<sub>set</sub>(Vo ALD), V<sub>set</sub>(Vo PVD), V<sub>reset</sub>(Cu ALD), V<sub>reset</sub>(Cu PVD) distributions of Cu/Ta<sub>x</sub>O<sub>y</sub>/Pt devices.



Fig.2  $R_{on}(Cu \text{ ALD})$ ,  $R_{on}(Cu \text{ PVD})$ ,  $R_{on}(V_O \text{ ALD})$ ,  $R_{on}(V_O \text{ PVD})$ distributions of Cu/Ta<sub>x</sub>O<sub>y</sub>/Pt devices.

References

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