

The Role of Heat Treatment in Enhanced Activity of Manganese Oxides for the Electrochemical Oxygen Reduction and Evolution

Yelena Gorlin,^a Dennis Nordlund,^b and Thomas F. Jaramillo^a

^aDept. of Chemical Engineering, Stanford University, 381 North-South Mall, Stanford, CA 94305 USA

^bSLAC National Accelerator Laboratory, 2575 Sand Hill Rd, Menlo Park, CA 94025 USA

Manganese oxides and oxyhydroxides (MnO_x) can exist in over 15 types of phases with Mn taking on 2⁺, 3⁺, 4⁺, and mixed valent oxidation states as well as various structural configurations.(1) Recently, an increasing number of phases has been linked to high activity for the oxygen reduction reaction (ORR)(2-6) and the oxygen evolution reaction (OER),(7-12) raising the importance of MnO_x catalysts to energy storage and conversion devices such as alkaline fuel cells, metal-air batteries, electrolysis cells, and photoelectrochemical hydrogen production. To accelerate development of MnO_x catalysts for these important technologies, it is necessary to determine the properties of active MnO_x surfaces under reaction conditions.

From thermodynamic considerations, the relevant ORR and OER bulk phases in alkaline conditions can be predicted as Mn₂O₃/MnOOH and MnO₂,(13) while the relevant ORR and OER surfaces have been recently proposed to be Mn₂O₃ covered with ½ monolayer of adsorbed OH and MnO₂ covered with adsorbed O, using density functional theory calculations.(14) These thermodynamically stable phases may or may not form in real catalysts because kinetic barriers may favour a different phase. Experimental characterization of MnO_x active surfaces in real catalysts under or after exposure to reaction conditions has so far been limited to a handful of studies. Under the ORR conditions, MnO_x has been shown to exist as Mn₃O₄,(15) Mn₂O₃,(16) and a mixture of Mn(III) oxide and MnO₂.(2, 17) Although from thermodynamic considerations the application of OER potentials is expected to produce MnO₂, exposure to more oxidative potentials relevant to the OER is most commonly associated with formation of a mixed oxide phase similar to birnessite-MnO₂, which has a Mn oxidation state of ~3.7-3.8.(9, 11, 15) Additionally, investigations that looked at the oxidation state of Mn as a function of applied potential in the region between the ORR and the OER found that the catalysts showed either a significant change in Mn oxidation state(2, 15) or a minor change that was not detected by the employed spectroscopic technique.(16) These seemingly conflicting reports indicate that the observed phase at the ORR and the OER potentials may differ from the expected thermodynamically stable phase calculated in a Pourbaix(13) or a surface Pourbaix diagram of MnO_x(14) and suggest that MnO_x surface at the ORR and the OER potentials may depend on the starting phase or preparation route.

In our previous study, we demonstrated that electrodeposition of MnO_x on glassy carbon followed by a heat treatment at 480°C leads formation of alpha-Mn₂O₃ phase with bifunctional activity for the ORR and the OER. Subsequent characterization of the ORR and the OER relevant surfaces using *ex-situ* XPS failed to reveal any differences in the Mn 2p_{1/2} satellite structure or the O 1s spectra of the samples relevant to the two reaction conditions, suggesting that heat treatment at 480°C limited the ability of MnO_x to change oxidation state to the very top surface layer, likely on the order of one monolayer.(16) The study discussed herein will focus on understanding the effect of heat treatment temperature on the pseudocapacitive behaviour of MnO_x catalysts and their activity for the ORR and the OER. In the study, we synthesize different types of MnO_x surfaces by electrodepositing MnO_x on glassy carbon (GC) and exposing the electrodes to five different heat treatment conditions: no heat treatment (“no ht”), heat treatment at 250°C, 350°C, 450°C, and 500°C. We then hold each of the five samples at an ORR relevant potential of 0.7 V or an OER relevant potential of 1.65 V and track the resulting oxidation state of Mn using surface sensitive *ex-situ* Mn L-edge XAS, a more sensitive probe for

distinguishing Mn oxidation state than XPS.(18) By interfacing the XAS spectra with electrochemical characterization we determine how the surface properties influence the activities for the ORR and the OER.

ACKNOWLEDGEMENTS

This material is based upon work supported as part of the Center on Nanostructuring for Efficient Energy Conversion (CNEEC) at Stanford University, an Energy Frontier Research Center funded by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences under Award Number DE-SC0001060. SEM and XPS characterization was performed at the Stanford Nanocharacterization Laboratory (SNL) part of the Stanford Nano Shared Facilities. XAS characterization was carried out at the Stanford Synchrotron Radiation Lightsource, a Directorate of SLAC National Accelerator Laboratory and an Office of Science User Facility operated for the U.S. Department of Energy Office of Science by Stanford University. The SSRL Structural Molecular Biology Program is supported by the DOE Office of Biological and Environmental Research, and by the National Institutes of Health, National Institute of General Medical Sciences (including P41GM103393)

REFERENCES

1. J. E. Post, *Proc. Natl. Acad. Sci. USA*, **96**, 3447 (1999).
2. F. H. B. Lima, M. L. Calegario and E. A. Ticianelli, *Electrochim. Acta*, **52**, 3732 (2007).
3. Y. L. Cao, H. X. Yang, X. P. Ai and L. F. Xiao, *J. Electroanal. Chem.*, **557**, 127 (2003).
4. Y. Gorlin and T. F. Jaramillo, *J. Am. Chem. Soc.*, **132**, 13612 (2010).
5. L. Q. Mao, T. Sotomura, K. Nakatsu, N. Koshiba, D. Zhang and T. Ohsaka, *J. Electrochem. Soc.*, **149**, A504 (2002).
6. F. Y. Cheng, Y. Su, J. Liang, Z. L. Tao and J. Chen, *Chem. Mater.*, **22**, 898 (2010).
7. M. Morita, C. Iwakura and H. Tamura, *Electrochim. Acta*, **24**, 357 (1979).
8. F. Jiao and H. Frei, *Chem. Commun.*, **46**, 2920 (2010).
9. R. K. Hocking, R. Brimblecombe, L. Y. Chang, A. Singh, M. H. Cheah, C. Glover, W. H. Casey and L. Spiccia, *Nature Chem.*, **3**, 461 (2011).
10. M. I. A. M. S. El-Deab, A. M. Mohammad, T. Ohsaka, *Electrochem. Commun.*, **9**, 2082 (2007).
11. I. Zaharieva, M. M. Najafpour, M. Wiechen, M. Haumann, P. Kurz and H. Dau, *Energy Environ. Sci.*, **4**, 2400 (2011).
12. A. Iyer, J. Del-Pilar, C. K. King'ondou, E. Kissel, H. F. Garces, H. Huang, A. M. El-Sawy, P. K. Dutta and S. L. Suib, *J. Phys. Chem. C*, **116**, 6474 (2012).
13. M. Pourbaix, *Atlas of Electrochemical Equilibria in Aqueous Solutions*, Pergamon Press (1966).
14. H.-Y. Su, Y. Gorlin, I. C. Man, F. Calle-Vallejo, J. K. Norskov, T. F. Jaramillo and J. Rossmeisl, *Phys. Chem. Chem. Phys.*, **14**, 14010 (2012).
15. Y. Gorlin, Lasalle-Kaiser, B.; Benck, J. D.; Sheraz, G.; Yachandra, V.; Yano, J.; Jaramillo, T.F., *J. Am. Chem. Soc.* accepted (2013).
16. Y. Gorlin and T. F. Jaramillo, *J. Electrochem. Soc.*, **159**, H782 (2012).
17. F. H. B. Lima, M. L. Calegario and E. A. Ticianelli, *J. Electroanal. Chem.*, **590**, 152 (2006).
18. Y. Gorlin, C.-J. Chung, D. Nordlund, B. M. Clemens and T. F. Jaramillo, *ACS Catal.*, **2**, 2687 (2012).