## Electrochemical Reactivity of Magnesium Ions With Sn-Based Binary Alloys

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Magnesium ion rechargeable battery systems receive considerable interest as candidate post-lithium systems. However, the exceedingly slow solid-state diffusion of the divalent  $Mg^{2+}$  cations has been a bottleneck in electrochemical insertion/extraction [1]. To date, there have been only a few kinds of anode materials such as Bi, Sb, Sn and Bi-Sb alloys [2,3]. Other potential anode materials such as alloy anodes, should be investigated as has been extensive in the case of lithium. In this paper we explored tin (Sn) alloys (Cu-Sn, In-Sn, Pb-Sn) as possible candidates for negative electrode materials.

All electrochemical experiments were performed in an Ar-filled glovebox (H<sub>2</sub>O, O<sub>2</sub> < 1ppm). A three-electrode electrochemical cell was employed using Mg rods as the counter and reference electrodes. Sn, Pb<sub>0.4</sub>Sn<sub>0.6</sub>, In<sub>0.5</sub>Sn<sub>0.5</sub>, Cu<sub>6</sub>Sn<sub>5</sub> and Cu<sub>3</sub>Sn were used as the working electrodes after cleaning with acetone and/or ethanol. A 0.95 mol dm<sup>-3</sup> solution of EtMgBr in THF (ca. 200ppm H<sub>2</sub>O) was used as the electrolyte.

Figure 1a displays cyclic voltammograms (CV) for pure Sn, where combination of reduction and oxidation waves were clearly observed. The redox peaks were observed at +0.14 V and +0.25 V *vs.* Mg, respectively, comparable with the potential of charge-discharge plateaus, where Sn +  $2Mg^{2+}$  + 4e =  $Mg_2Sn$  is suggested [3]. The maximal current density was as large as about 12  $\mu$ A cm<sup>-2</sup>.

Insertion/desertion of Mg ions were also successful in In-Sn and Pb-Sn alloys. Fig. 1b shows the CV for In<sub>0.5</sub>Sn<sub>0.5</sub> alloy which consist of In<sub>3</sub>Sn and In<sub>0.2</sub>Sn<sub>0.8</sub>, showed broad redox peaks, possibly indicating that both In<sub>3</sub>Sn and In<sub>0.2</sub>Sn<sub>0.8</sub> phases are active against Mg. Shown in Fig. 1c is the case of eutectic Pb-Sn alloy, several reduction peaks were seen (0.04 V, 0.08 V, and 0.13 V vs. Mg), while one oxidation peak was seen (+0.29 V vs. Mg at 10th cycle). Since Pb-Sn system is a typical eutectic alloy where Pb-rich and Sn-rich phases coexist, the several cathodic peaks imply that both Sn-rich and Pbrich phases can react electrochemically to form such as Mg<sub>2</sub>Sn and Mg<sub>2</sub>Pb. The rather asymmetric CV curve may imply that, while Pb-rich and Sn-rich phase react with Mg individually, the Mg dissolution undergo in a cooperative way to reform Pb-Sn alloy: it is possible that the reformation of Pb-Sn alloy is a driving force for the demagnesiation process.

Figure 2 displays the CV for Sn, In-Sn and Pb-Sn at 10th cycle. The redox peaks of In-Sn and Pb-Sn alloys reached as large as about 130  $\mu$ A cm<sup>-2</sup>, ten times larger than that of elemental Sn. Here, taking into account that the melting points of In-Sn and Pb-Sn alloys (about 120 °C and 180 °C, respectively) are much lower than that of elemental Sn (232 °C), alloys with lower melting points can be advantageous in terms of Mg diffusion rate.

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Figure 1. CV data for (a) Sn, (b) Pb-Sn, and (c) In-Sn electrodes cycled between +0.05 V vs. Mg and the open circuit potential.



**Figure 2**. CV data of the 10th cycle for the Sn, In-Sn, and Pb-Sn electrodes.