

First-Principles Study of Oxygen Vacancy and Hydrogen Impurity Effects in the Pseudo-hexagonal Ta₂O₅

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Resistance random access memory (ReRAM) is one of the most promising candidates for future non-volatile memory devices to replace conventional flash memory. ReRAM has attracted much attention because of its simple structure, high speed operation, low power consumption, and high density integration.¹ Among the candidates for resistive materials for ReRAM applications, conventional binary transition metal oxide (TMO) is particularly attractive because they can be easily mass produced. In particular, Ta₂O₅-based ReRAM devices showed superior endurance properties and switching performances.² The resistance switching behavior is controlled by the applied voltage or current pulse at the electrodes. Lattice defects in the bulk oxide or near the interface with the electrode are currently believed to play an important role in the switching between the resistive off state to a more conductive on state. Based on experimental observations, several resistance switching models have been suggested so far for TMO-based devices.³ However, the underlying principles of the switching mechanism are still lacking a detailed understanding. In this study, the structures and energies of Ta₂O₅ were calculated using density-functional theory. The electronic interactions are described within GGA+*U* formalism, where on-site Coulomb corrections are applied on 5*d* orbital electrons of Ta atoms (*U^d*) and 2*p* orbital electrons of O atoms (*U^p*). The results are improved when the correlation correction is introduced additionally on O 2*p* orbitals by employing GGA+*U^d*+*U^p* approach. Using this combined approach, based on the bulk properties, as structural stability, crystal structure symmetry, and band-gap, the pseudo-hexagonal referenced as “asymmetric” by previous authors⁴ was found to be a quite stable polymorph of Ta₂O₅ and proposed as a new candidate for resistive switching applications. The supercell structure (2x2x4) of the pseudo-hexagonal Ta₂O₅ is illustrated in Fig. 1(a). Fig. 1(b) shows x-ray diffraction (XRD) pattern of Ta₂O₅ film and diffraction data calculated by GGA+*U^d*+*U^p* in the pseudo-hexagonal Ta₂O₅. The diffraction data of the pseudo-hexagonal Ta₂O₅ are comparable to the experimental result. Both the experimental result and diffraction data of the orthorhombic *cmmm* and triclinic *p1* Ta₂O₅ are not consistent with one another. All our three proposed structures are much more stable than δ - and β -Ta₂O₅. Not only the band-gap energies of δ - and β -Ta₂O₅ calculated by GGA+*U^d*+*U^p* are in good agreement with those of PBE0 by Nashed *et al.*,⁵ but also the band-gap energies of all our three proposed structures obtained by GGA+*U^d*+*U^p* agree well with the experimental band-gap energy. It is well known that the oxygen vacancies play an important role in the electrical conductivity of Ta₂O₅. To simulate the oxygen vacancy in the pseudo-hexagonal Ta₂O₅, we employed GGA+*U^d*+*U^p* in the calculation of the electronic structure of 2x2x4 supercell incorporating one oxygen vacancy. The total density of states calculated in the pseudo-hexagonal Ta₂O₅ with one

neutral oxygen vacancy is shown in Fig. 2(a). The oxygen vacancy causes a defect state in the band gap. This electron-occupied state is situated at 2.5 eV above the top of the valence band. This is also in good agreement with the experimental x-ray photoelectron spectroscopy spectrum from a previous report.⁶ The partial density of states of Ta and O atoms in the same structure are shown in Figs. 2(b) and 2(c). The defect state generated by the oxygen vacancy is formed mostly by 5*d* orbitals of Ta atoms with a small contribution of O 2*p* states. Furthermore, the effects of the hydrogen impurity on the electronic structures were also investigated, as shown in Fig. 3. The hydrogen impurity extends the defect state region in the band gap and is found to be preferentially located in the neighborhood of the oxygen vacancy. The hydrogen impurity also reduces the oxygen vacancy formation energy, which is predicted to have an influence on lowering forming voltage.

References

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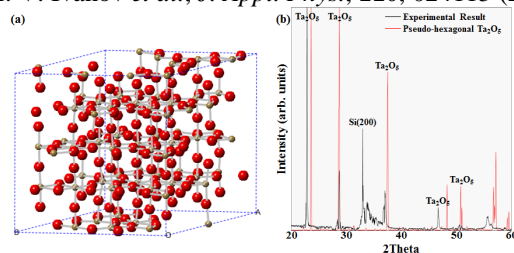


Fig. 1. (a) Structure of the pseudo-hexagonal Ta₂O₅. Small yellow ball denotes Ta atom and big red one O atom. (b) XRD pattern of Ta₂O₅ film and diffraction data calculated in the pseudo-hexagonal Ta₂O₅.

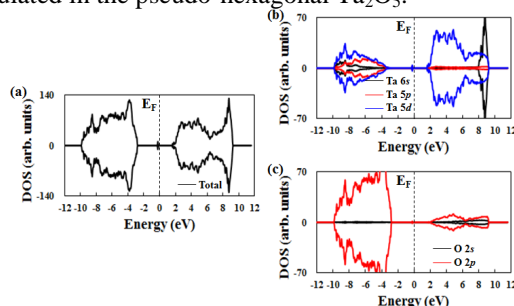


Fig. 2. (a) Total density of states and partial density of states of (b) Ta and (c) O atoms calculated in the pseudo-hexagonal Ta₂O₅ with one neutral oxygen vacancy.

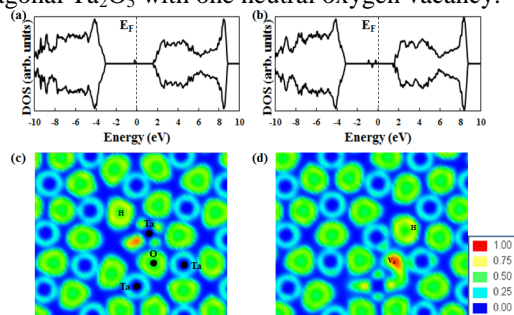


Fig. 3. (a-b) Total density of states and (c-d) electron localization function (ELF) calculated in the pseudo-hexagonal Ta₂O₅ (a, c) without/(b, d) with one neutral oxygen vacancy and with one hydrogen impurity. ELF: showing the missing electrons in the blue region and the highly localized ones in the red region.