Synthesis and Electrochemical Characterization of Spinel-type Mg Complex Oxides as Positive Electrode Active Materials for Mg Batteries

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Mg batteries have been attracting significant interest as low-cost batteries with high electromotive force, high capacity, and safety, because Mg has the lowest standard electrode potential among the air-stable metals, and Mg resources are abundant and widely spread in the world compared to Li resources. However, the difficulties of diffusion and desolvation of divalent Mg<sup>2</sup> ions limit the choice of positive electrode (cathode) active materials. Factually, only Chevrel compounds are the workable positive electrode active materials with a relatively high charge/discharge rate and high reversibility at this time.<sup>1</sup> Their redox potentials are about 1.2 V vs. Mg, which are far below the anodic stability limit of electrolytes being developed recently (around 3 V vs. Thus, the discovery of positive electrode active Mg).<sup>2</sup> materials with higher redox potentials immediately leads to the enhancement of the electromotive force.

In the present work, we synthesized spinel-type Mg complex oxides, MgM<sub>2</sub>O<sub>4</sub> (M: transition metal, e.g. Cr, Mn, Fe, and Co) via inverse coprecipitation.<sup>3</sup> First, metallic salt solutions were prepared by dissolving 0.080 M Mg(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O; 0.160 M M(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O (M: Co, Mn) or M(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O (M: Cr, Fe) in deionized water. A sodium carbonate solution was then prepared as a precipitation solution for pH control. The metallic salt solutions were added dropwise to the precipitation solution. The resulted suspensions were vigorously stirred at 70-80 °C for 30 min and then filtered. The filtered precipitates (precursors) were rinsed with deionized water at 80 °C, followed by drying at 80 °C for 24 h. The precursors were pulverized by planetary boll milling at 500 rpm for 8 x 15 min at intervals of 5 min. The pulverized precursors were calcined at 500 or 750 °C for 24 h. The structure parameters of the obtained spineltype oxide powders, specifically the occupancies of atomic sites were analyzed on the basis of the synchrotron XRD data using the Rietveld program RIETAN-FP.<sup>4</sup>

Fig. 1 shows the crystal structure of a spinel type oxide AB<sub>2</sub>O<sub>4</sub>. In normal spinel oxides AB<sub>2</sub>O<sub>4</sub>, all the divalent cations, A, occupy the tetrahedral sites (8a sites) and all the trivalent cations, B, occupy the octahedral sites (16d sites). In inverse spinel oxides, all the A cations and half the B cations occupy octahedral sites (16d sites) and the remaining half of the B cations occupy tetrahedral sites (8a sites). There are also disordered intermediate states with random A and B cation distributions. In this abstract, we show the result of  $MgCo_2O_4$  as an example. Fig. 2 presents the synchrotron XRD profile of the spineltype MgCo<sub>2</sub>O<sub>4</sub> powder calcined at 500 °C. Rietveld refinement revealed that the synthesized MgCo<sub>2</sub>O<sub>4</sub> was a disordered spinel, (Mg<sub>0.53</sub>Co<sub>0.47</sub>)(Mg<sub>0.24</sub>Co<sub>0.76</sub>)<sub>2</sub>O<sub>4</sub>. The expected charge and discharge reactions of MgCo2O4 are as follows:

Charge: MgCo<sub>2</sub>O<sub>4</sub>
$$\rightarrow$$
xMg<sup>2+</sup>+Mg<sub>1-x</sub>Co<sub>2</sub>O<sub>4</sub>+2xe  
(2xCo(III) $\rightarrow$ 2xCo(IV)) } }(1)

Discharge:  $xMg^{2+}+Mg_{1-x}Co_2O_4+2xe \rightarrow MgCo_2O_4$ (2 $xCo(IV) \rightarrow 2xCo(III)$ ) (2)

Fig. 3 shows the cyclic voltammogram measured at 0.1 mV s<sup>-1</sup> with 3-electrode setup using  $MgCo_2O_4$  as the working electrode (WE) and Mg ribbons as the counter electrode (CE) and reference electrode (RE). Two characteristic redox peak couples  $(Red_1/Ox_1 and$ Red<sub>2</sub>/Ox<sub>2</sub>) are observed at around 3 and 2 V vs. Mg, respectively. It should be noted that the electrolyte was oxidatively decomposed above 2.7 V vs. Mg,<sup>5</sup> and there is an offset of anodic current density above 2.7 V. Although further detailed investigation is required to determine which redox peak couple corresponds to Eqs.1-2, it is expected that the valence change of Co(III)/Co(IV) occurs at higher potentials. Considering this fact, the redox peak couple around 2 V vs. Mg (Red<sub>2</sub>/Ox<sub>2</sub>) would correspond to the valence change of Co(II)/Co(III), implying the existence of the oxygen vacancies in the structure of MgCo<sub>2</sub>O<sub>4</sub> because of charge compensation.



Fig. 1 Crystal structure of a spinel type oxide  $AB_2O_4$ .



Fig. 2 Synchrotron XRD profile of  $MgCo_2O_4$  powder calcined at 500 °C for 24 h.



**Fig. 3** Cyclic voltammogram measured at 0.1 mV s<sup>-1</sup> using  $MgCo_2O_4$  as the working electrode in THF solution containing 0.50 M PhMgCl and 0.25 M AlCl<sub>3</sub>.

References

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