Electrode Performance of Lithium Containing Layered Sodium Iron Manganese Oxides for Rechargeable Na-Ion Batteries

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Introduction

Rechargeable sodium batteries are the promising candidate for large-scale applications because the sodium resources are unlimited everywhere. Recently, our group reported that P2-type Na2/3 [Fe1/2Mn1/2]O2 consisting of abundant elements in natural resources provides 190 mAh g^{-1} of reversible capacity in the sodium cell[1]. This reversible capacity is higher than those of $O3-NaFeO_2$ and O'3-NaMnO₂ electrodes. However, the P2-type phase is generally known to be a sodium deficient phase, which leads to an excess discharge capacity at the initial cycle. Our group also reported that P2-type Na_{5/6}[Li_{1/4}Mn_{3/4}]O₂, containing a higher amount of sodium as P2-phase, delivers a larger initial charge capacity compared to P2- $Na_{2/3}[Fe_{1/2}Mn_{1/2}]O_2\ phase[2].$ In this study, to increase the initial charge capacity and further increase a reversible capacity, the solid solution of $Na_{2/3}[Fe_{1/2}Mn_{1/2}]O_2$ – Na_{5/6}[Li_{1/4}Mn_{3/4}]O₂ is synthesized, and its structural and electrochemical properties are studied.

Experimental

Solid solution between Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂ and Na_{5/6}[Li_{1/4}Mn_{3/4}]O₂ was synthesized by a solid-state reaction. Na₂CO₃, Li₂CO₃, MnCO₃·H₂O and α -Fe₂O₃ were mixed by mechanical milling, pelletized and then heated at 700 – 900°C for 12 hours in air. Crystal structure and morphology of the samples were examined by using powder synchrotron X-ray diffraction measurements and scanning electron microscopy. Electrochemical properties were tested using coin-type cell. Positive electrodes consisted of the active material, acetylene black and polyvinylidenefluoride (PVdF) with a gravimetric ratio of 80:10:10. Metallic sodium was used as a negative electrode. The electrolyte used was 1.0 mol dm⁻³ NaClO₄ / PC : FEC (98 : 2)[3].

Results and Discussion

Figure 1 shows the synchrotron X-ray diffraction (XRD) patterns of $xNa_{2/3}[Fe_{1/2}Mn_{1/2}]O_2$ (1-x) $Na_{5/6}[Li_{1/4}Mn_{3/4}]O_2$ (x = 0, 1/3, 1/2, 2/3 and 1.0). All Bragg diffraction lines, except a superlattice line at ca. 7 degrees, are indexed based on the P2-type layered structure with a space group $P6_3/mmc$, indicating the of solid formation the solution between Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂ Na_{5/6}[Li_{1/4}Mn_{3/4}]O₂. and As increasing a portion of Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂, a peak profile Therefore, the existence of of 10l lines broadens. stacking faults is expected to be significant, especially for the sample of x = 2/3. Initial charge and discharge curves for two end-members of P2-type Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂ and $Na_{5/6}[Li_{1/4}Mn_{3/4}]O_2$, and the one-to-one solid solution (x = 0.5), are shown in Fig. 2. For the sample of x = 0.5, the excess discharge capacity is effectively decreased compared with that of Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂, and higher reversible capacity of approximately 200 mAh g⁻¹ than those of two end-members is achieved. A characteristic initial charge plateau, similar to Li_2MnO_3 -based high capacity electrode materials, is observed as increase in the lithium contents. The partial oxygen loss is, therefore, expected similar to the observation for Li_2MnO_3 -based materials[4]. From these results, the relationship between the crystal structures and electrochemical properties will be further discussed.

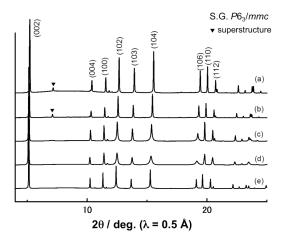


Fig. 1 Synchrotron XRD patterns of the solid solution samples of $xNa_{2/3}[Fe_{1/2}Mn_{1/2}]O_2 - (1-x)Na_{5/6}[Li_{1/4}Mn_{3/4}]O_2$ for x = (a) 0, (b) 1/3, (c) 1/2, (d) 2/3, and (e) 1.0. The end-members of P2-Na_{5/6}[Li_{1/4}Mn_{3/4}]O_2 (x = 0) and Na_{2/3}[Fe_{1/2}Mn_{1/2}]O₂ (x = 1) were synthesized at 700 and 900 °C, respectively. The samples of x = 1/3, 1/2 and 2/3 were synthesized at 800 °C.

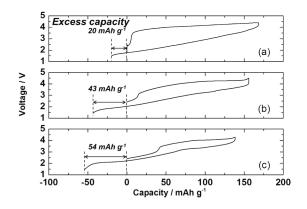


Fig. 2 Initial charge and discharge curves of Na cells using the solid solution samples of $xNa_{2/3}[Fe_{1/2}Mn_{1/2}]O_2 - (1-x)Na_{5/6}[Li_{1/4}Mn_{3/4}]O_2$ for x = (a) 0, (b) 1/2 and (c) 1.0 at C/20 rate.

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