

## Electrochemical Reduction of CO<sub>2</sub> using Supported Cu<sub>2</sub>O Nanoparticles

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The oxidation state of Cu is reported to have a significant effect on its product selectivity during CO<sub>2</sub> electroreduction. For metallic Cu, it is widely accepted that CO and HCOOH are the main products formed at low overpotentials, while hydrocarbons (notably CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub>) become the dominant products at higher potentials [1,2]. However, a few groups have reported CH<sub>3</sub>OH as a major product using pre-oxidized Cu electrodes [3] or supported Cu<sub>2</sub>O nanoparticles [4]. It has also been reported that supported Cu<sub>2</sub>O layers, although being reduced to Cu metal under CO<sub>2</sub> reduction conditions, result in a decreased overpotential for CO formation, relative to Cu metal [5].

In this study we examine the stability and product selectivity of CO<sub>2</sub> electroreduction using colloidally prepared Cu<sub>2</sub>O nanoparticles [4]. The starting solution of aqueous CuCl<sub>2</sub> with polyethylene glycol (PEG) surfactant is reduced using L-ascorbic acid sodium (LAAS), aged for 6 hours, then rinsed with deionized water. The resulting slurry is mixed with a Nafion/ethanol solution, sonicated for 10 minutes, and dried onto Toray carbon paper. The CO<sub>2</sub> reduction experiments are performed using a two compartment cell with a Nafion membrane separator, a Pt wire counterelectrode, and Ag/AgCl reference electrode. The working electrolyte is aqueous 0.5M KHCO<sub>3</sub> that is bubbled continuously with CO<sub>2</sub>. Product concentrations are determined using gas chromatography. Gas phase products (H<sub>2</sub>, CO, CH<sub>4</sub>, and C<sub>2</sub>H<sub>4</sub>) are sampled directly from the CO<sub>2</sub> purge gas leaving the reactor. Liquid products (mainly CH<sub>2</sub>H<sub>5</sub>OH) are measured by taking syringe samples of the electrolyte at 15 minute intervals.

The SEM images of freshly prepared electrodes show the cubic morphology expected for Cu<sub>2</sub>O particles, as reported by Chang *et al.* [4]. X-ray diffraction measurements confirm that only peaks assigned to the Cu<sub>2</sub>O phase are present (i.e., no peaks assignable to metallic Cu are observed). After being used for CO<sub>2</sub> reduction (i.e., 120 minutes at -1.745 V(SCE)), the XRD measurements indicate that nearly all of the Cu<sub>2</sub>O is reduced to Cu metal. The SEM images show the reduced electrode material retains the same cubic morphology as the starting Cu<sub>2</sub>O nanoparticles, but the surface of the cubes become decorated with much smaller clusters.

The CO<sub>2</sub> reduction experiments are performed at a single potential, -1.745 V(SCE). This potential is selected to be well into the region for CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub> formation on Cu electrodes [1,6]. For our supported Cu<sub>2</sub>O nanoparticle electrodes at this potential, the largest CO<sub>2</sub> reduction product is C<sub>2</sub>H<sub>4</sub>, with a Faradaic efficiency of 34%. We also observe the formation of C<sub>2</sub>H<sub>5</sub>OH in parallel to C<sub>2</sub>H<sub>4</sub>, with a Faradaic efficiency of about 6%. In contrast, the formation rate of CH<sub>4</sub> is much lower, with a Faradaic efficiency of only 1%. (The largest overall product in these experiments is H<sub>2</sub>, which we attribute to H<sub>2</sub>O reduction on the incompletely covered carbon support. Control experiments using both unmodified and Nafion-treated carbon paper confirm very low activity for CO<sub>2</sub> reduction in the absence of Cu<sub>2</sub>O nanoparticles on these supports.)

The most noticeable feature of the CO<sub>2</sub> reduction behavior using the supported Cu<sub>2</sub>O nanoparticle electrodes appears to be the large C<sub>2</sub>H<sub>4</sub>/CH<sub>4</sub> product ratio. At our selected electrode potential of -1.75 V(SCE), Hori *et al.* [6] reported that CH<sub>4</sub> formation actually exceeded C<sub>2</sub>H<sub>4</sub> formation by a factor of 2 on polycrystalline Cu. Using single crystal studies, the group later showed the C<sub>2</sub>H<sub>4</sub>/CH<sub>4</sub> ratio is sensitive to surface structure, with C<sub>2</sub>H<sub>4</sub> formation being favored on higher index surfaces [7]. The highest ratio they reported (i.e., observed using the Cu(711) surface) was C<sub>2</sub>H<sub>4</sub>:CH<sub>4</sub> = 14:1, which approaches the value reported here. More recently, Schouten *et al.* [8] were able to resolve separate pathways for C<sub>2</sub>H<sub>4</sub> and CH<sub>4</sub> formation on Cu(111) and Cu(100) surfaces, with the latter (somewhat) more open surface providing an independent, lower energy pathway for C<sub>2</sub>H<sub>4</sub> formation.

Our present results support a model in which the Cu<sub>2</sub>O particles are converted to metallic Cu during CO<sub>2</sub> reduction. The morphological evolution of the starting Cu<sub>2</sub>O nanoparticles during this reduction leads to the formation of more highly dispersed Cu clusters on the surface of the converted particles. The dispersed Cu clusters are expected to contain a higher concentration of more open crystal faces and lower co-ordination surface atoms, which leads to the observation of a higher C<sub>2</sub>H<sub>4</sub>/CH<sub>4</sub> ratio, relative to low-index planar Cu surfaces. Additional work is underway to further characterize these dispersed Cu clusters, and to explore methods to improve their stability during prolonged electrochemical operation.

### References

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